

# Structural characterization of tin doped indium oxide films prepared by magnetron sputtering

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The structural characteristics of transparent and conducting thin films of tin doped indium oxide (ITO) have been studied by X-ray diffraction and electron microscopy. The ITO films have been prepared by radio frequency sputtering employing a planar magnetron configuration. The effects of variation of substrate temperature and annealing have been studied. Films deposited at substrate temperatures up to 230° C show a marked  $\langle 110 \rangle$  preferred orientation, whereas those deposited at 330° C are preferentially oriented in the  $\langle 311 \rangle$  direction. Transmission electron micrographs of ITO films deposited on KCl crystals show the possible effects of strain on the structure, which are minimized to a great extent by annealing.

## 1. Introduction

Films of indium oxide and tin doped indium oxide (ITO) have been used as transparent conducting layers in many sophisticated devices such as solar cells, solar photothermal conversion devices, liquid crystal displays, etc. Extensive research has been done and is still being carried out for the improvement of electrical and optical properties of these films [1]. The main stress in research has been on the improvement of sheet resistance and transmittance of the films. These properties are also dependent on the structure of the film. Although several studies of ITO films report on some of the structural aspects [2-12], a concerted, systematic study of structural characterization does not appear to exist.

In an earlier paper [13] we reported the results of our studies on the electronic and optical properties of indium tin oxide (ITO) films prepared by the radio frequency (r.f.) magnetron sputtering method.

In this paper we have reported the results of

our studies on the structural aspects of ITO films by X-ray diffraction and transmission electron microscopy. Since in our earlier study the properties of ITO films were found to be most sensitive to the substrate temperature, we have mainly varied this parameter in our structural studies.

## 2. Experimental procedures

ITO films were deposited by r.f. sputtering employing a planar magnetron configuration (CVC Inc., USA). The target employed was a disc of hot pressed powder (8 inch diameter, 1/4 inch thick) with a composition of 90%  $\text{In}_2\text{O}_3$  and 10%  $\text{SnO}_2$  by weight (supplied by CERAC Inc., USA). The r.f. power supplied to the target was 200 W. Unless specifically mentioned, the argon pressure employed was  $4.5 \times 10^{-3}$  torr. The other experimental details are described in [13].

For the X-ray diffraction studies, the films were deposited on Corning 7059 glass substrate. The deposition rate was of the order of



10 nm min<sup>-1</sup>. The films were prepared at substrate temperatures of 50, 230 and 330° C. Films in the thickness range 600 to 850 nm were studied. X-ray line profiles were recorded using a Philips Geiger Counter Diffractometer (PW1050, 1051) with monochromatic CuK $\alpha_1$  radiation from a Philips X-ray generator (PW1130). The line profiles were chart recorded at a speed of 1° per minute. All the measurements were carried out at room temperature (300 K).

Transmission electron microscopy (Hitachi HU-11F model operating at 75 kV) was performed on a few samples deposited on KCl single crystals and glass. The thickness of the ITO films was in the range 15 to 40 nm (required for Transmission Electron Microscopy) which is much lower than that of the films used for X-ray diffraction. Both bright field imaging and selected area diffraction studies were done to supplement the X-ray diffraction data on the structural aspects of the ITO films.

The thicknesses of the films was measured by a stylus type thickness measuring apparatus (Planer Products, UK). The dark conductivity has been measured by the van der Pauw technique and Hall effect has been studied by the four probe method with an electromagnet having a maximum field strength of 7.6 kG.

### 3. Results and discussion

X-ray diffraction measurements have been carried out on as-deposited ITO films and also on films annealed for one hour in a vacuum at 350° C and in air at 400° C. Typical results of X-ray diffraction measurements have been summarized in Table I. Intensities have been reported in terms of peak heights. In order to check the effects of line broadening, the line areas of the first and second most intense peaks have also been calculated and compared with the results in terms of peak heights (Table II). Both give virtually the same result.

All the lines in the diffraction patterns can be assigned to the cubic Tl<sub>2</sub>O<sub>3</sub> (bixbyte) structure of In<sub>2</sub>O<sub>3</sub> [14]. There is no evidence of the presence of a crystalline phase of SnO<sub>2</sub> in the X-ray diffractometer traces in any of the films examined. It may be mentioned here that Lehman and Widmer [6] could not detect the presence of SnO<sub>2</sub> in co-sputtered ITO films from X-ray diffraction traces, up to 40 mol % SnO<sub>2</sub> concentra-

TABLE II The relative intensity  $I/I_{\max}$ (%) and the relative area  $A/A_{\max}$ (%) of X-ray profiles from ITO films (two most intense peaks considered)

Film ITO	<i>hkl</i>	$I/I_{\max}$ (%)	$A/A_{\max}$ (%)
Deposited at 50° C	(440)	100	100
	(400)	94	96
Deposited at 230° C	(440)	100	100
	(222)	91	96
Deposited at 330° C	(622)	100	100
	(440)	45	47

tion. Fan *et al.* [5] and Itoyama [7] have also reported the absence of SnO<sub>2</sub> or Sn<sub>3</sub>O<sub>4</sub> peaks in the diffraction patterns of their r.f. sputtered films.

The texture of a film depends upon the surface free energy and the most preferred alignment is of the set of planes with minimum surface free energy. The actual alignment depends strongly upon the deposition conditions and in our case changes substantially with a change in the substrate temperature.

For films deposited at 50° C, the two peaks (440) and (400) appear prominently, indicating the existence of both  $\langle 110 \rangle$  and  $\langle 100 \rangle$  textures. The (622) peak is third in order of intensity (Fig. 1a). Itoyama [7], for his as-deposited magnetron sputtered ITO films (substrate temperature  $\sim 140$  to  $180^\circ$  C) obtained an amorphous-like X-ray diffraction pattern. Only on annealing in air or in N<sub>2</sub> at 400° C, the ITO films became crystalline. In the present work, the samples, even those that were as-deposited, show well-developed crystallinity.

These films, when annealed in vacuum, show no remarkable change. However, when annealed in air for one hour at 400° C, the films also develop a  $\langle 111 \rangle$  texture as shown by the presence of (222) peak. The intensity of the (440), (400) and (622) peaks also increase, indicating enhanced crystallinity (Fig. 1b). This is also corroborated by Hall effect measurements. For a film deposited at 50° C, the room temperature mobility is about 29 cm<sup>2</sup> V<sup>-1</sup> sec<sup>-1</sup>. On annealing in air at 400° C for 1 h, the mobility is found to be doubled ( $\sim 58$  cm<sup>2</sup> V<sup>-1</sup> sec<sup>-1</sup>), which could be attributed to an increased crystallinity.

For the films deposited at 230° C, the (440) peak is still the most prominent one (Fig. 2). However, the intensity of the (400) peak diminishes and that of the (222) peak increases,

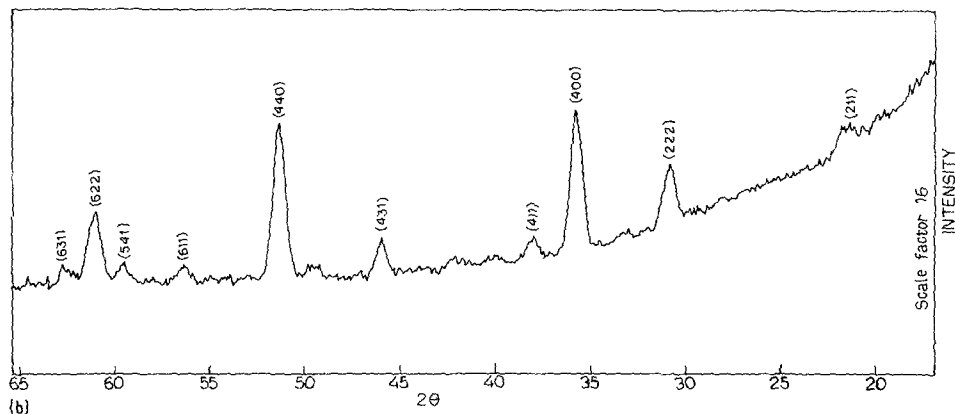
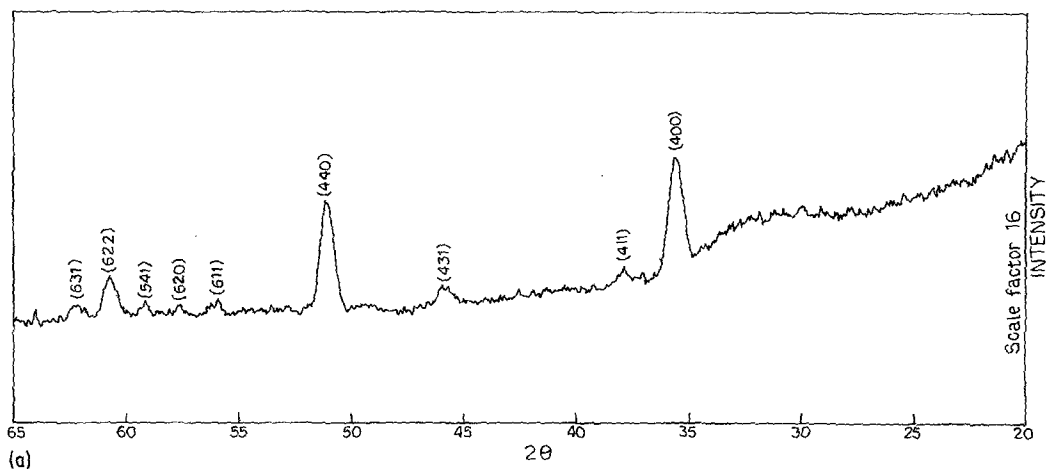


Figure 1 X-ray diffraction patterns for  $\text{In}_2\text{O}_3:\text{Sn}$  films: (a) deposited at  $50^\circ\text{C}$ ; (b) deposited at  $50^\circ\text{C}$  and annealed in air at  $400^\circ\text{C}$ .

showing a  $\langle 111 \rangle$  preferred orientation. The (622) peak also is more prominent. In the case of films deposited at  $330^\circ\text{C}$ , more high angle peaks are detectable compared with films deposited at lower temperatures (Fig. 3). The (440) peak diminishes and the most prominent peak is (622), showing a  $\langle 311 \rangle$  texture.

Overall, there is an increase in the degree of crystallinity of ITO films with increasing substrate temperature. The resistivity also decreases from  $3.1 \times 10^{-4}$  to  $7.4 \times 10^{-5} \Omega\text{cm}$  as the substrate temperature is increased from 50 to  $330^\circ\text{C}$ . As reported in our earlier paper [13], there is an increasing trend in the conductivity,

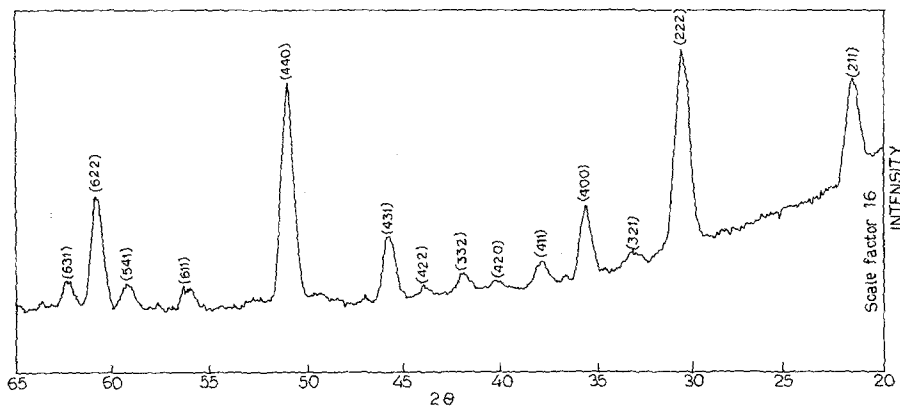


Figure 2 X-ray diffraction pattern for  $\text{In}_2\text{O}_3:\text{Sn}$  film deposited at  $230^\circ\text{C}$ .

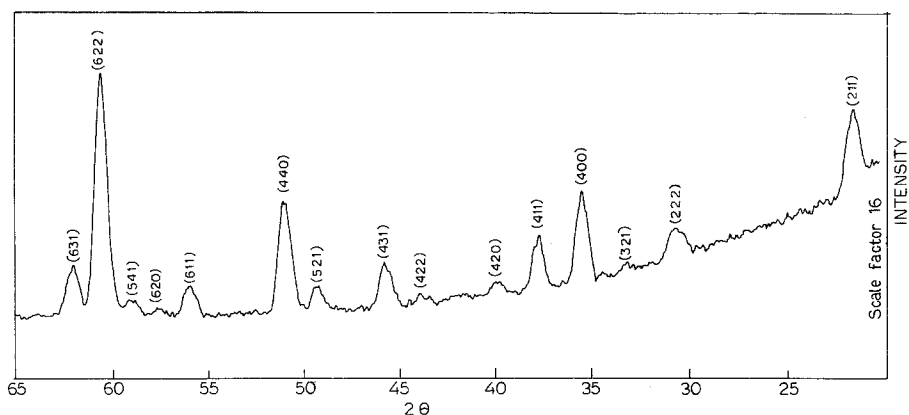


Figure 3 X-ray diffraction pattern for  $\text{In}_2\text{O}_3$ : Sn film deposited at  $330^\circ\text{C}$ .

carrier concentration and mobility with increase in the substrate temperature for deposition.

The effect of annealing in air at  $400^\circ\text{C}$ , for ITO films deposited at  $230$  and  $330^\circ\text{C}$  is not quite discernible in the X-ray profiles as in case of films deposited at  $50^\circ\text{C}$  (Table I). Perhaps, with a higher annealing temperature the change could have been prominent.

It may be mentioned here that X-ray diffraction studies on ITO films deposited by different sputtering techniques revealed either a  $\langle 111 \rangle$  or  $\langle 100 \rangle$  preferred orientation, depending upon the deposition parameters and also post-deposition annealing [2–9].

The lattice constant,  $a$ , of the ITO films increases with the deposition temperature, though not linearly. For films deposited at  $50^\circ\text{C}$ ,  $a = 1.010\text{ nm}$ ; for those deposited at  $230^\circ\text{C}$ ,  $a = 1.011\text{ nm}$ ; and for films deposited at  $330^\circ\text{C}$ ,  $a = 1.012\text{ nm}$ . The value for  $a$  from ASTM data for  $\text{In}_2\text{O}_3$  powder is  $1.0118\text{ nm}$ . Kulaszewicz [15] has also reported an increase in

the lattice constant of ITO films, prepared by spray pyrolysis, as the spraying temperature of the substrate is increased.

As a supplement to the X-ray studies, the effects of deposition temperature on the microstructure and crystallinity of ITO films were studied by transmission electron microscopy. The as-deposited ITO films prepared at  $50^\circ\text{C}$  do not show very well developed crystallinity (Fig. 4). The films consist of poorly crystalline regions of the size  $\sim 10$  to  $20\text{ nm}$ . With the increase in substrate temperature there is a gradual increase in this size up to  $\sim 20$ – $50\text{ nm}$ . Fig. 5b shows the typical micrograph for a film deposited on a substrate at  $330^\circ\text{C}$ . The selected area diffraction pattern however, still contains only a few slightly diffuse rings (Fig. 5a). It may be mentioned here that Nath *et al.* [11] reported TEM studies on  $\text{In}_2\text{O}_3$  (Sn) films and observed that with the increase in deposition temperature, the crystallite size also increased.

The apparent difference between the degree of

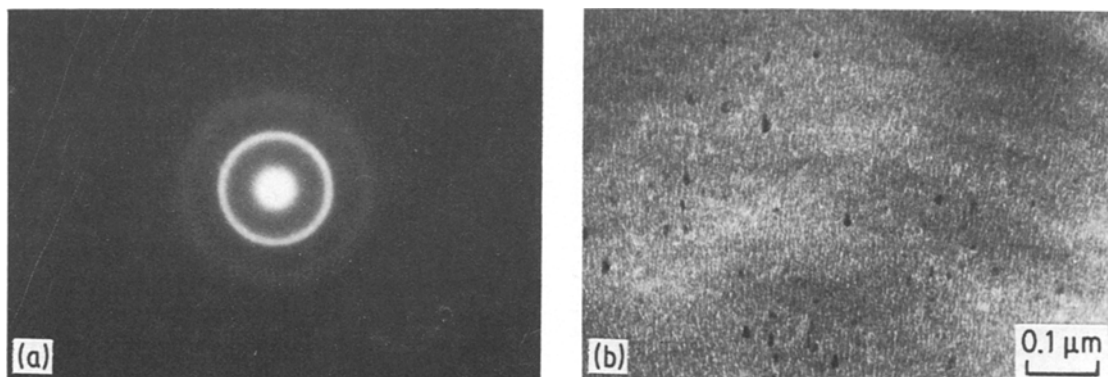


Figure 4 (a) Electron diffraction pattern and (b) TEM micrograph for  $\text{In}_2\text{O}_3$ : Sn film deposited at  $350^\circ\text{C}$ .

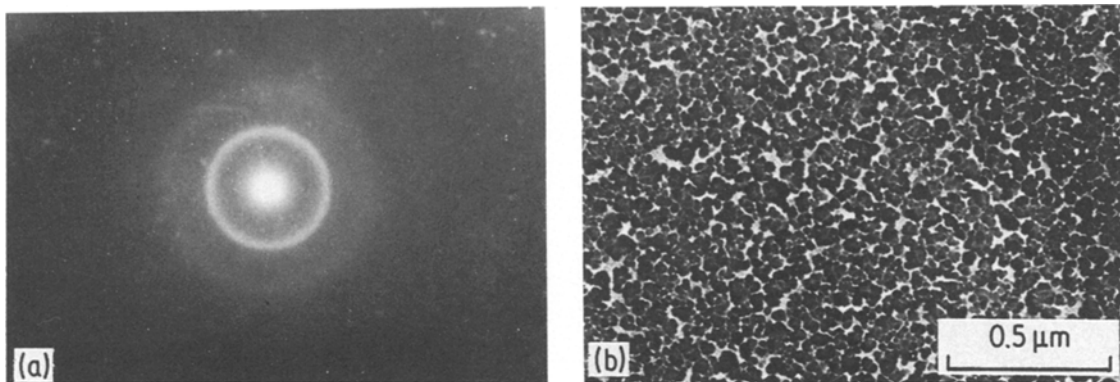


Figure 5 (a) Electron diffraction pattern and (b) TEM micrograph for  $\text{In}_2\text{O}_3:\text{Sn}$  film deposited at  $330^\circ\text{C}$ .

crystallinity of ITO films observed by the X-ray diffraction technique and TEM studies may be attributed to strain effects. These effects could arise due to the small thickness (15 to 40 nm) of the films used for TEM studies. The effect of substrate was checked by depositing ITO on glass at  $50^\circ\text{C}$ . TEM studies showed no significant change in these samples than those deposited on KCl.

To probe further into the results obtained by transmission electron microscopy, the effects of annealing on the ITO films were studied. For films deposited at  $50^\circ\text{C}$ , the selected area diffraction patterns show remarkable recovery effect on annealing both in vacuum and in air (Figs. 6a and 7a). There is no apparent increase in the crystallite size under vacuum annealing, yet the strain effects are removed. As a check, the resistivity of 15 nm thick ITO film deposited at  $50^\circ\text{C}$  was measured. The resistivity of the as-deposited film was  $5.6 \times 10^{-4} \Omega\text{cm}$ . After annealing in

vacuum at  $350^\circ\text{C}$  for one hour, the resistivity decreased to  $2.1 \times 10^{-4} \Omega\text{cm}$ . A thicker film (140 nm) did not show any change in resistivity after similar treatment. The decrease in the resistivity of the thin film on vacuum annealing, could be attributed to the removal of strain effects, which are not prominent in thicker films. When annealed in air on the other hand, micrographs of films deposited at  $50^\circ\text{C}$  reveal regions of large grains  $\sim 150\text{ nm}$ , with a good degree of crystallinity (Fig. 7b). The resistivity of the sample increases by one order of magnitude on air annealing. This is due to the diffusion of oxygen into the ITO film, which reduces the concentration of oxygen vacancies. However, the mobility increases, as already mentioned in the discussion for X-ray diffraction studies. For films deposited at higher substrate temperatures, vacuum annealing does not show any prominent recovery effect. Annealing in air however shows significant recrystallization of the films (Fig. 8).

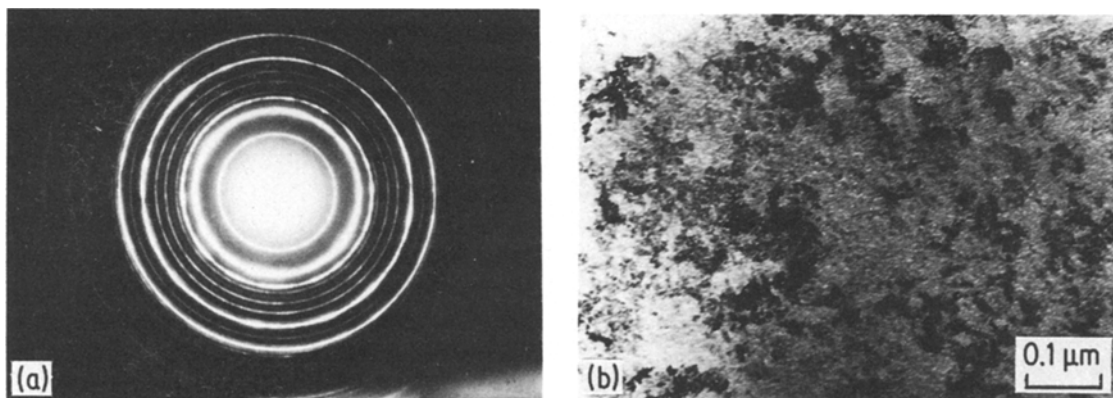


Figure 6 (a) Electron diffraction pattern and (b) TEM micrograph for  $\text{In}_2\text{O}_3:\text{Sn}$  film deposited at  $50^\circ\text{C}$  and annealed in vacuum at  $350^\circ\text{C}$ .

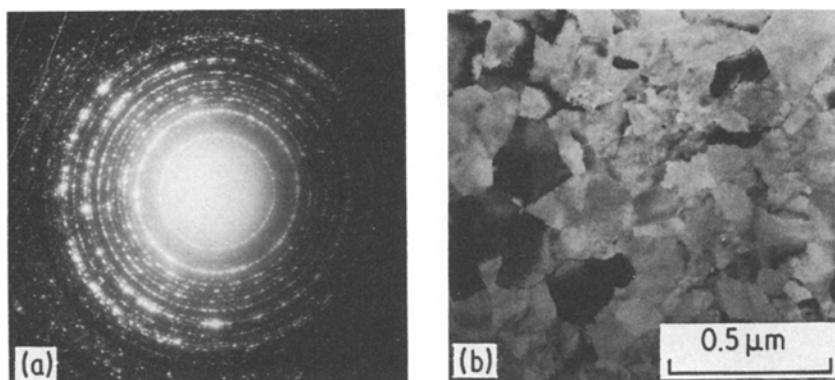


Figure 7 (a) Electron diffraction pattern and (b) TEM micrograph for  $\text{In}_2\text{O}_3:\text{Sn}$  film deposited at  $50^\circ\text{C}$  and annealed in air at  $400^\circ\text{C}$ .

To check whether strain effects are minimized by a slower rate of deposition, TEM studies were performed on ITO samples prepared at high argon pressure ( $12.5 \times 10^{-3}$  torr) during sputtering. The substrate temperature was  $230^\circ\text{C}$ . As the argon pressure was increased from 4.5 to  $12.5 \times 10^{-3}$  torr during sputtering, the rate of deposition dropped to about one third of the previous value. (Therefore the TEM samples were prepared at argon pressure of  $12.5 \times 10^{-3}$  torr). The selected area diffraction pattern shows excellent crystalline structure even for the as-deposited films (Fig. 9). In contrast, as already stated, for films deposited at  $4.5 \times 10^{-3}$  torr, good crystalline structure is obtained only after annealing. Thus, slower rate of deposition minimizes strain effects and improves the microstructure considerably.

#### 4. Conclusions

A gradual change from  $\langle 110 \rangle$  to  $\langle 311 \rangle$  texture

is observed in X-ray profiles for ITO films prepared on glass by r.f. magnetron sputtering as the deposition temperature is increased from 50 to  $330^\circ\text{C}$ . Films deposited at  $50^\circ\text{C}$  show better crystallinity when annealed in air at  $400^\circ\text{C}$ . Films deposited at higher substrate temperatures ( $230^\circ\text{C}$ ,  $330^\circ\text{C}$ ) do not show prominent annealing effects.

Transmission electron microscopy studies on much thinner specimens on KCl crystals show a different microstructural characteristic which does not change with higher deposition temperature. Annealing treatment under different environment has a significant effect on the structure of these films. Slower rate of deposition at increased argon pressure during sputtering produces good crystalline structure even without annealing.

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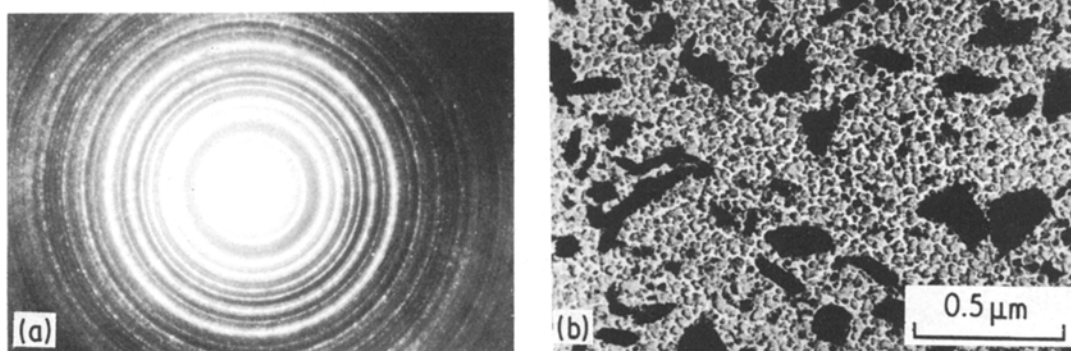


Figure 8 (a) Electron diffraction pattern and (b) TEM micrograph for  $\text{In}_2\text{O}_3:\text{Sn}$  film deposited at  $330^\circ\text{C}$  and annealed in air at  $400^\circ\text{C}$ .

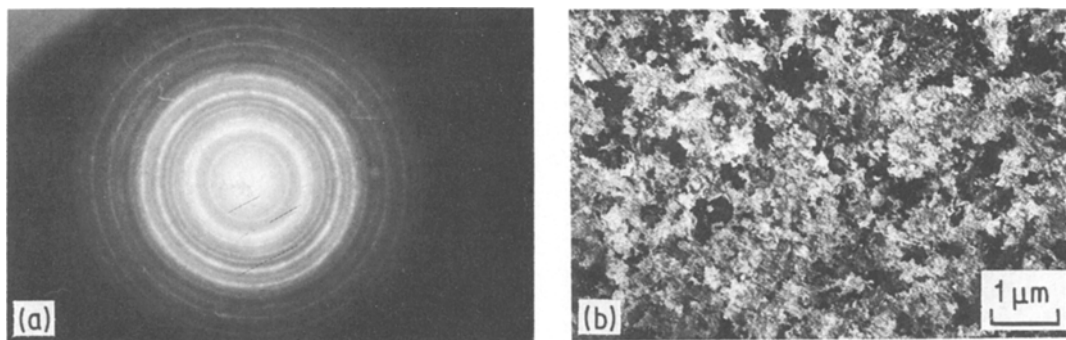


Figure 9 (a) Electron diffraction pattern and (b) TEM micrograph for  $\text{In}_2\text{O}_3 \cdot \text{Sn}$  film deposited at high argon pressure ( $12.5 \times 10^{-3}$  torr).

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## References

1. K. L. CHOPRA, S. MAJOR and D. K. PANDYA, *Thin Solid Films* **102** (1983) 1.
2. J. L. VOSSER, *RCA Rev.* **32** (1971) 289.
3. D. B. FRASER and H. D. COOK, *J. Electrochem. Soc.* **119** (1972) 1368.
4. M. HECQ, A. DUBOIS and J. VAN CAKENBERGHE, *Thin Solid Films* **18** (1973) 117.
5. J. C. C. FAN and F. J. BACHNER, *J. Electrochem. Soc.* **122** (1975) 1719.
6. H. W. LEHMANN and R. WIDMER, *Thin Solid Films* **27** (1975) 359.
7. K. ITOYAMA, *Jpn. J. Appl. Phys.* **17** (1978) 1191.
8. M. FUJINAKA and A. A. BEREZIN, *Thin Solid Films* **101** (1983) 7.
9. A. J. STECKL and G. MOHAMMED, *J. Appl. Phys.* **51** (1980) 3890.
10. P. NATH and R. F. BUNSHAH, *Thin Solid Films* **69** (1980) 63.
11. P. NATH, R. F. BUNSHAH, B. M. BASOL and O. M. STAFFSUD, *ibid.* **72** (1980) 463.
12. A. P. MAMMANA, E. S. BRAGA, I. TORRIANI and R. L. ANDERSON, *ibid.* **85** (1981) 355.
13. SWATI RAY, RATNABALI BANERJEE, N. BASU, A. K. BATABYAL and A. K. BARUA, *J. Appl. Phys.* **54** (1983) 3497.
14. ASTM Powder Diffraction File, No. 6-0416 (1955).
15. S. KULASZEWICS, *Thin Solid Films* **76** (1981) 89.

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